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Elucidation of the CO₂ adsorption mechanism of $[Zn_2(mtz)_2(ox)]$ using neutron

powder diffraction

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ABSTRACT

Metal-organic framework $[Zn_2(tz)_2(ox)]$ (CALF-20) has attracted great attention due to its excellent ability to capture carbon dioxide. There are great interests to develop similar adsorbents for gas adsorption and separation. To develop more efficient porous adsorbent, it is essential to study the relationship between these structures and properties. Neutron diffraction has been proved to be an excellent tool for determining both the structural details of MOF host and the precise locations of adsorbed gas within the pore, offering unique opportunities for understanding the structure-properties relationship. Herein, we report the synthesis and structure characterization of MOF $[Zn_2(mtz)_2(ox)]$, which exhibits high CO₂ adsorption capacity. Neutron powder diffraction experiment on the solvated, the activated and CO₂ loaded samples unveils the preferred binding sites of CO₂ within the MOFs, where CO₂ locates toward the center of the pore and interacts with methyl group or triazole *via* C-H···O hydrogen bonding. The adsorption process of CO₂ in $[Zn_2(mtz)_2(ox)]$ is accompanied by the cell volume expansion, so $[Zn_2(mtz)_2(ox)]$ with more compact structure can show a better adsorption performance. The structure-properties relationship in $[Zn_2(mtz)_2(ox)]$ elucidated by present study offer a path to develop more advanced porous physisorbent materials.

Metal-organic frameworks (MOFs) have gained great interest due to their potential applications in catalysis [1-3], gas adsorption and separation [4-6], and drug delivery [7-9]. MOFs are usually assembled through solvothermal reactions with various metal ions and organic linkers, leading to structural diversity and functional adjustability [10-13]. In general, the properties and pore size of these porous materials can be adjusted for specific applications by using different metal ions or organic ligands. For instance, when using different metal ions, MAF-123-Cd showed highest usable storage capacity of C_2H_2 compared with MAF-123-Zn and MAF-123-Mn [14]. Alternatively, by substitution of organic ligands with different lengths, the pore sizes as well as performances of MOFs can also be adjusted, for example, the effective window size of [Cu(bpy-n)₂(SiF₆)] (bpy-1 = 4,4'-bipyridine; bpy-2 = 1,2-bis (4 pyridyl) ethane) is *ca*. 8 and 10.6 Å, respectively. At room temperature and ambient pressure, [Cu(bpy-1)₂(SiF₆)] exhibits a higher CO₂ capacity (23.1 wt%) and CO₂/CH₄ selectivity (10.5) than [Cu(bpy-2)₂(SiF₆)] [15].

Recently, MOF $[Zn_2(tz)_2(ox)]$ (CALF-20) has attracted great attention due to its excellent ability to capture carbon dioxide, leading to the emergence of many similar adsorbents, such as Zn-ox-mtz and NCU-20 [16-20]. It is interesting that NCU-20 and CALF-20 exhibit significantly different properties despite of similar crystal structures, coordination modes and unit formulas. The different properties might originate from delicate changes in the unit cell and atomic positions accompanied by the variation of structures and pore sizes. Therefore, exploring the structure-properties relationship of these MOFs is of great significance for developing new MOF adsorbents. However, most of these MOF samples cannot retain in single crystal when the solvent is removed, so that it is difficult to characterize the crystal structure by single crystal X-ray diffraction (SXRD). Moreover, the detection of hydrogen atom position is always a problem for X-rays.

Neutron bears the merit of high sensitivity to light atom and good penetration, and has been proved to be an essential tool for determining both the structural details of MOF host and the locations of adsorbed gas molecules [21-23]. This work reports the synthesis and structure characterization of $[Zn_2(mtz)_2(ox)]$, which exhibits good CO₂ adsorption capacity. Neutron powder diffraction (NPD) experiments were performed on the solvated, the activated and CO₂ loaded samples. The strong incoherent neutron scattering to hydrogen sometimes causes high background in NPD data, but in our NPD experiment, the incoherent neutron scattering is rather weak due to a small atomic fraction of hydrogen in $[Zn_2(mtz)_2(ox)]$. With the reasonable-background NPD data, we have precisely determined the structural transformation as well as the interactions between CO₂ and the host framework, which was corroborated by the Grand Canonical Monte Carlo simulations. As a result, we finally unveiled the CO₂ adsorption mechanism and structure-properties relationship of $[Zn_2(mtz)_2(ox)]$.

 $[Zn_2(mtz)_2(ox)]$ was prepared according to the reported method with slight modifications [17]. ZnCO₃ (250 mg), oxalic acid (187 mg), 3-methyl-1*H*-1,2,4-triazole (830 mg), H₂O (12 mL) and *n*-BuOH (4 mL) was added in a 50 mL Teflon autoclave. The mixture was vigorously stirred for 30 min at room temperature and heated in an oven at 453 K for 3 days. The obtained white precipitate was separated by centrifuge, washed several times with methanol, and dried under vacuum.

Sorption isotherms for CO₂ (99.99%, from Mulai air) was measured at 298 K on the ASAP 2020 instrument. The sorbent $[Zn_2(mtz)_2(ox)]$ was activated at 373 K for 8 h under dynamic vacuum before sorption measurements.

X-ray powder diffraction (PXRD) of $[Zn_2(mtz)_2(ox)]$ -solv were measured at room temperature using a Rigaku Smartlab powder diffractometer. Neutron powder diffraction (NPD) data of the solvated ($[Zn_2(mtz)_2(ox)]$ -solv), the activated ($[Zn_2(mtz)_2(ox)]$), and the CO₂ loaded samples ($[Zn_2(mtz)_2(ox)]$ -0.28CO₂) were collected using the time-of-flight neutron powder diffractometer, iMATERIA, at the Materials and Life Science Experimental Facility of the Japan Proton Accelerator Research Complex (J-PARC) and the time-of-flight high-resolution neutron diffractometer (TREND) at China Spallation Neutron Source (CSNS). Activation of the samples was carried out at 373 K for 8 h in vacuum prior to the neutron experiments. CO₂ was loaded using a gas handling system at 1 bar and 298 K for 0.5 h. The data collecting was performed after all the gas had been adsorbed, allowing sufficient time to achieve thermal equilibrium. All the NPD data were collected at room temperature.

Rietveld refinements on the NPD patterns of the MOFs were performed using the TOPAS software package [24], in which the initial framework model was geometrically constrained. The CO₂ guests were treated as a rigid body. The mass center, orientation, and

occupancies of the guest molecule were first refined, followed by the refinements comprised all free structural variables from both the framework and guest molecules.

GCMC simulations were performed to generate CO₂ positions in the MOFs using the Sorption module in Material Studio software package. The structure of $[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$ obtained from the NPD experiment were used as the starting configuration. The solvent and gas molecules were removed from the framework prior to the simulations, denoted as $[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$ (removed), which is analogous to the activation in the experiment. Both MOF framework and CO₂ molecules were regarded as rigid bodies. The adsorption site simulation was performed at 298 K and 1.0 bar by the fixed loading task and Metropolis method. The atomic partial charges of the host MOF framework and guest CO₂ molecules were obtained by the QEq method. The framework-CO₂ interaction and the gas-gas interaction were taken from the universal force field (UFF). The cut-off radius used for the Lennard-Jones interactions is 15.5 Å and the electrostatic interactions were calculated based on the Ewald summation method. Each simulation consists of an equilibration period of 5.0×10^6 iterations followed by a production run of further 1.0×10^7 iterations.

Both PXRD and NPD confirms that the synthesized sample ($[Zn_2(mtz)_2(ox)]$ -solv) is of high purity and the crystal structure is a isomer of Zn-ox-mtz (Figs. 1a and Fig. 2a) [17].

 $[Zn_2(mtz)_2(ox)]$ -solv crystallizes into a three-dimensional framework structure with the monoclinic space group $P2_1/c$ (Fig. 1b). The Zn(II) ion is coordinated by two O-donors from oxalate group [Zn-O = 2.162(6) and 1.965(3) Å] and three N-donors from bridging triazole units [Zn-N = 2.061(9), 2.038(6) and 2.077(9) Å]. The coordination induces a one-dimensional channel made of zig–zag segment running through the framework, which contains uncoordinated water (1.0 per Zn). So the chemical formula for the as-synthesized material is $[Zn_2(mtz)_2(ox)] \cdot 2H_2O$. Although the space group and unit formula of $[Zn_2(mtz)_2(ox)]$ -solv resemble those of Zn-ox-mtz solv [17], the structure of $[Zn_2(mtz)_2(ox)]$ -solv differs slightly. For instance, due to the rotation of triazole ring accompanied by the shift of atomic positions in $[Zn_2(mtz)_2(ox)]$ -solv, the pore aperture for $[Zn_2(mtz)_2(ox)]$ -solv is 7.00 × 3.40 Å², which is significantly larger than 5.30 × 3.50 Å² found in Zn-ox-mtz-solv [17]. There are also small differences in the lattice parameters between $[Zn_2(mtz)_2(ox)]$ -solv and Zn-ox-mtz-solv (Table 1) [17]. In $[Zn_2(mtz)_2(ox)]$ -solv, the *a*, *b* and β expand while the *c* contracts compared with Zn-ox-mtz-solv. As a result, the cell volume of $[Zn_2(mtz)_2(ox)]$ -solv is larger that of Zn-ox-mtz-solv.



Fig.1 (a) PXRD patterns of [Zn2(mtz)2(ox)] solv, (b) Three-dimensional pillared-layer frameworks of [Zn2(mtz)2(ox)] solv (for clarity, solvents were removed).



Fig. 2 NPD patterns, structure and \angle NZnN angle of MOFs: (a, d, g) [Zn₂(mtz)₂(ox)] ·solv, (b, e, h) [Zn₂(mtz)₂(ox)] and (c, f, i) [Zn₂(mtz)₂(ox)] ·0.28CO₂.

Table 1 Lattice parameters of Zn-ox-mtz solv	/ [17	$[Zn_2(mtz)_2(ox)] \cdot so$	v, [Zn2(mtz)2(ox)] and	$[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$	obtained from NPD
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Isomer	Space group	a (Å)	<i>b</i> (Å)	c (Å)	α (°)	β (°)	γ (°)	$V(Å^3)$	
$Zn\text{-}ox\text{-}mtz\text{\cdot}solv^{[17]}$	$P2_{1}/c$	8.6166(2)	7.8948(1)	10.5066(2)	90	96.97(2)	90	709.45(2)	
$[Zn_2(mtz)_2(ox)]$ ·solv	$P2_{1}/c$	8.6881(1)	8.0947(1)	10.4423(2)	90	98.51(1)	90	726.31(1)	
$[Zn_2(mtz)_2(ox)]$	$P2_{1}/c$	8.7207(1)	8.2027(1)	10.3689(1)	90	99.58(1)	90	731.38(1)	

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(1) 8 3420(1) 10 2752(1) 90 100.83(1) 90 73734(1)			
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The CO₂ adsorption capacity of desolvated $[Zn_2(mtz)_2(ox)]$ was obtained through CO₂ sorption experiments at 298 K in the pressure range of 0.0-1.0 bar. It can be seen from Fig. 3 that $[Zn_2(mtz)_2(ox)]$ exhibits Type-I behavior with reversible uptake. The maximum CO₂ uptake of $[Zn_2(mtz)_2(ox)]$ is 66.9 cm³/g (2.99 mmol/g) at 298 K and 1.0 bar, which is slightly lower to Zn-ox-mtz (68.8 cm³/g or ~3.07 mmol/g).



Fig. 3. CO₂ adsorption isotherms for [Zn₂(mtz)₂(ox)] at 298 K.

To determine the structure of the framework and the interaction between the adsorbed CO₂ molecules and the framework, the solvated $([Zn_2(mtz)_2(ox)] \cdot solv)$, the activated $([Zn_2(mtz)_2(ox)])$, and the CO₂ loaded samples $([Zn_2(mtz)_2(ox)] \cdot 0.28CO_2)$ were studied using NPD (Figs. 2a-f). Compared to $[Zn_2(mtz)_2(ox)] \cdot solv$, the cell volume of $[Zn_2(mtz)_2(ox)]$ expands by 0.69% while that of $[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$ expands by 1.5% (Table 1). The volume expansions are due to the rotation of one of the triazole rings in the frameworks that elongates lattice constant *a* and *b*. The $\angle NZnN$ angle also increases during the rotation of the ring, which is 92.29°, 94.19° and 96.09° in $[Zn_2(mtz)_2(ox)] \cdot solv$, $[Zn_2(mtz)_2(ox)]$ and $[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$, respectively (Figs. 2g-i). Therefore, the MOFs' structures show the dynamic response of the framework to external stimuli such as heat or guest molecules. Moreover, since all CALF-20 analogues feature the triazole rings, such ring rotation might occur in CALF-20 analogues as well.

NPD studies on CO₂ loaded [Zn₂(mtz)₂(ox)] also determined the location of the adsorbed gas molecule, which resides toward the center of the pore (Site I), as shown in Figs. 4a and b. The CO₂ molecule not only interacts with the triazole forming three C-H···O hydrogen bonds (H-O lengths: 2.578, 2.702 and 3.040 Å) but also with three methyl groups to form five C-H···O hydrogen bonds (H-O lengths: 2.593, 2.709, 3.404, 3.048, and 3.096 Å). The occupancies for these CO₂ molecules at Site I refined to values of 0.14, yielding the overall formula [Zn₂(mtz)₂(ox)]·0.28CO₂ for the CO₂ loaded [Zn₂(mtz)₂(ox)] samples. The value is lower than the experimental uptake from the CO₂ isotherm at 298 K, which gives the overall formula of [Zn₂(mtz)₂(ox)] 1.14CO₂ (Fig. 3). The discrepancy can be attributed to the presence of disordered CO₂ molecules in the cavity, as indicated in nuclear density maps of CO₂ loaded [Zn₂(mtz)₂(ox)] data obtained from our Maximum Entropy Method (MEM) analysis. Nevertheless, this discrepancy does not prevent us from determining the binding mechanisms. To compare the CO₂ adsorption sites determined computationally and experimentally, the experimental CO₂ positions are superimposed on three-dimensional adsorption density distribution maps for CO₂ generated by GCMC simulations in Fig. 4c. The symmetry of the probability clouds suggests one distinct adsorption sites, which is consistent with the experiment result. In light of the adsorption density distribution, the favorable adsorption site of CO₂ is in excellent agreement with the experimentally observed position.

Site I determined experimentally gives a profound insight into the effect of the structure transformation on the CO₂ adsorption capacity. Since Site I locates near the center of the pore, the strength of its interaction with the framework (C-H···O bonds) depends on the volume of the framework. As a result, a more compact framework facilitates stronger host-guest interaction, which triggers higher CO₂ adsorption capacity. This is exactly what happens in $[Zn_2(mtz)_2(ox)]$ and Zn-ox-mtz. The CO₂ sorption isotherm of Zn-ox-mtz [17] exhibits higher CO₂ uptake than that of isomorphous $[Zn_2(mtz)_2(ox)]$, whereas the cell volume of Zn-ox-mtz solv is smaller than that of $[Zn_2(mtz)_2(ox)]$. Since the two isomers were obtained by only different solvents, the cell volume as well as the CO₂ adsorption capacity of similar MOFs can be tuned by varying the solvents.



Fig. 4 (a) Site I in $[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$ for CO₂. (b) Side view of Site I in $[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$ for CO₂. (c) Adsorption density distribution calculated by GCMC simulation at 298 K and 1.0 bar and Site I in $[Zn_2(mtz)_2(ox)] \cdot 0.28CO_2$ for CO₂.

In conclusion, NPD studies of $[Zn_2(mtz)_2(ox)] \cdot solv$, $[Zn_2(mtz)_2(ox)]$ and CO₂ loaded $[Zn_2(mtz)_2(ox)]$ revealed, $[Zn_2(mtz)_2(ox)]$ exhibits delicate softness to external stimuli. The expansion of $[Zn_2(mtz)_2(ox)]$ during the gas loading was attributed to the rotation of triazole rings. In the molecular level, preferred binding site within the pores of $[Zn_2(mtz)_2(ox)]$ determined by NPD was in good agreement with the results of GCMC simulations, *i.e.*, CO₂ locates toward the center of the pore *via* C-H···O hydrogen bonding

interactions with methyl group or triazole. As a result, $[Zn_2(mtz)_2(ox)]$ analogues with more compact structure bear stronger hydrogen bonding interaction, inducing better adsorption performance. The primary cell volumes as well as the CO₂ adsorption capacity of similar MOFs can be tuned by varying the solvents in the synthesis.

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Declaration of competing interest

The authors declare no competing interests.

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Graphical Abstract



Neutron powder diffraction experiments on the solvated, the activated and CO_2 loaded $[Zn_2(mtz)_2(ox)]$ unveils the preferred binding sites of CO_2 within the MOFs as well as the delicate structural transformation in the adsorption process.

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Declaration of Interest Statement

☑ The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

□ The author is an Editorial Board Member/Editor-in-Chief/Associate Editor/Guest Editor for this journal and was not involved in the editorial review or the decision to publish this article.

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